AFS RESEARCH REPORT

COMPARISON OF MONOTONIC AND CYCLIC PROPERTIES OF DUCTILE IRONS IN THE AFS/DOE STRAIN-LIFE FATIGUE DATABASE FOR CAST IRON

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Abstract

A strain-life fatigue database for cast irons was developed utilizing American Foundry Society (AFS) and the United States Department of Energy (DOE) funding. The database contains monotonic and cyclic property data, as well as the associated chemical analysis and microstructural data for a variety of cast irons, including gray, ductile, compacted graphite, and white cast irons.

This paper first reviews the contents and format of the database. The database and its associated report contain no comparative analysis of any of the cast iron grades. The fresh approach of this paper is to analyze and compare the data across each section size and strength-ductility combination for four ductile iron grades, i.e., ferritic 60-40-18, ferritic-pearlitic 65-45-12, pearlitic 100-70-03 and tempered martensitic (Q&T) 120-90-02.

The ductile cast iron results in the database generally exhibited the expected trends. Modulus and Poisson's ratio were the same for all the conditions. Monotonic strength always decreased with increasing elongation and ferrite content. The higher strength grades and conditions exhibited greater high cycle fatigue resistance and decreased low cycle fatigue resistance.

Heat treatments had a mixed effect on monotonic properties. Annealing decreased the monotonic strength of 60-40-18 whereas normalizing increased the strength and dramatically increased fatigue resistance of pearlitic 100-70-03. Monotonic ductility was only slightly affected by heat treatment.

However, in most conditions, the heat treatments produced greater low cycle fatigue lives for high ductility conditions. Although the high cycle fatigue resistance was affected less by heat treatment, slightly increased high cycle fatigue lives were obtained with higher strength conditions. The normalized 25 mm samples of grade 100-70-03 exhibited significantly greater monotonic strength as well as fatigue resistance in both the low and high cycle regime.

Keywords: strain-life fatigue, low cycle fatigue, cast iron, ductile cast iron, microstructure, heat treatment

Introduction

Description of Fatigue

A standard definition¹ of fatigue is that it is the process of progressive, localized, and permanent structural change occurring in a material subjected to conditions

- that produce fluctuating stresses and strains at some point or points, and
- that may culminate in cracks or complete fracture after a sufficient number of fluctuations.

Another common definition² introduces the concepts that

- fatigue fractures occur at fluctuating stresses with a maximum value less than the ultimate tensile strength of the material,
- fatigue failure generally occurs at loads which applied statically would produce little perceptible effect, and

• fatigue fractures begin as minute cracks that grow under the action of fluctuating stress.

High versus Low Cycle Fatigue

For over hundred years, the stress-life or S-N method was the first approach used in an attempt to quantify and design by fatigue. Although the S-N approach can be used in design applications where the applied stress is primarily within the elastic range of material response, the strain-life approach is required for low cycle fatigue life predictions between 10 and 100,000 cycles.³

The low cycle or strain-life approach also offers the advantage that both fatigue stress and strain are tested, analyzed, and modeled. Although high cycle fatigue only handles stress, low cycle or strain-life fatigue testing offers the opportunity for stress and strain to be calculated from each other. It is nearly impossible to measure stress with load cells in complex and moving structures, whereas strain can be readily measured with optical or strain gauge methods. Another key difference between high and low cycle fatigue is that high cycle fatigue is strength and crack-initiation controlled where stronger materials exhibit greater fatigue lives in the high cycle range. High cycle fatigue testing commonly produces an endurance or fatigue limit (strength) at long lives (10 million cycles). The fatigue limit for p %survival is defined as the limiting value of fatigue strength for p %survival as N becomes very large (like 10 million cycles), where p may be any number, such as 95, 90, and so forth.¹ Low cycle fatigue on the other hand considers the regime where ductility and crack propagation dominate life such that more ductile materials exhibit greater fatigue lives in the low cycle range.

For several reasons, bending fatigue strength has been a property that was measured historically. Now, modern designers require low cycle or strain-life fatigue properties of cast iron to populate their computer-aided design databases so that cast iron can be compared to compendia⁴ of strain-life properties of steels and other competing materials. Strain-life fatigue properties can not be determined in bending and must be determined axially so that the cyclic stress and strain can be related.

Another important distinction is that modern fatigue databases do not contain fatigue limits, which are essentially the stresses below which materials are expected to survive fatigue. Rather, finite element modeling programs must be able to predict lives and locations of fatigue fractures in the vicinity of holes, thickness changes, and other stress concentrators. The cast iron strain-life fatigue database discussed in this paper offers designers the opportunity to perform such calculations with cast iron.

Motivation for the Database

Strain-life fatigue testing was conducted on a variety of cast irons for a variety of reasons. Such fatigue data is intended for use in current endurance modeling methods. Thus, designers can select optimal materials and design geometries to foster

- (a) lighter products,
- (b) energy efficient operations,
- (c) longer and more predictable product service life, and
- (d) more energy efficient product manufacture.

Database Description

Multiple grades of a variety of cast irons were subjected to a considerable variety of

tests, and a database containing the strain-life fatigue properties of multiple grades of cast iron was developed and published⁵ in 2003. The work was originally funded by the U.S. Department of Energy/Cast Metals Coalition (DOE/ CMC)⁶ with guidance by the American Foundry Society (AFS). Since the publication of the original DOE report,⁷ additional grades have been tested and the database contents have been augmented on an ongoing basis with additional AFS research funds. AFS has also developed an on-line materials database searching tool called CADS, which will eventually include this strain-life fatigue database product. For information see: http://www.metalcastingvirtuallibrary.com/cads/cads.aspx

Table 1 shows the current contents of the database based on testing by the author.⁵ The database also contains contributed datasets from other sources, but these results are not included in the analysis conducted for this paper. Test samples for most of the grades were obtained from Y-blocks in the as-cast condition, but a few grades were tested with heat treated Y-blocks or with specimens machined from cast parts.

 Table 1. Materials Tested at Element (as of March 2012)

 The conditions marked in yellow were selected for analysis in this paper.

Auster	mpered Ductile Iron: ASTM A 8	97-06		
ASTM Grade	Material Condition	Section Size		
110 / 70 / 11		48 mm by 43 mm by 185 mm		
110770711		section of rectangular casting		
125 - 80 - 10				
(130 / 90 / 09)	Austompored			
150 - 100 - 7	Austempered	25 mm V block		
(150 / 110 / 07)		25 mm Y-DIOCK		
175 – 125 – 4				
200 - 155 - 1				
Duc	tile Iron: ASTM A 536-84 (Re 19	99)		
ASTM Grade	Material Condition	Section Size		
	As-cast	25 mm Keel block		
60 40 40	Subcritical Anneal	25 mm Keel block		
60 - 40 - 18	Full Anneal	25 mm Y-block		
	Heavy Section; As-Cast	76 mm Y-block		
65 – 45 - 12	As-cast	25 mm Keel block		
	As-cast	25 mm Keel block		
100 - 70 - 03		25 mm Keel block		
	Normalized	76 mm Y-block		
120 - 90 - 02	Quenched and Tempered	25 mm Keel block		
Compacted	d Graphite Iron: ASTM A 842-85	5(Re 1997)		
ASTM Grade	Material Condition	Section Size		
300				
350	As-Cast	25 mm Y-block		
	As-Cast	25 mm Y-block		
400	As-Cast	76 mm Y-block		
	empered Ductile Iron: ASTM A 897-06 Material Condition Section Size 48 mm by 43 mm by 185 mm section of rectangular casting Austempered 25 mm Y-block ctile Iron: ASTM A 536-84 (Re 1999) Material Condition Section Size As-cast 25 mm Keel block Subcritical Anneal 25 mm Keel block Full Anneal 25 mm Keel block As-cast 25 mm Keel block Quenched and Tempered 25 mm Keel block Quenched and Tempered 25 mm Y-block As-Cast 25 mm Y-block			
ASTM Grade	Material Condition	Section Size		
20B	As-Cast	25 mm Y-block		
	As-Cast	13 mm Round		
30B	As-Cast	25 mm Y-block		
000	As-Cast	76 mm Y-block		
35B	As-Cast	25 mm Y-block		
000	As-Cast with $C E = 3.7$	25 mm Y-block		
40B	As-Cast with Normal C E =	23 mm T-block		
466	4 0	25 mm Y-block		
Austompo	red Grav Iron: No Standard Spo	cification		
Base Iron	Material Condition	Section Size		
Class 30B	Austempered	25 mm V-block		
Abraci	on-Resistant Cast Iron: ASTM	A 532		
ASTM Grade	Material Condition	Section Size		
		0601011 0126		
(25% Cr)	Quenched & tempered	16 mm rectangular bars		
(207001)				

Experimental Procedures

This section of this paper is highly abbreviated versus the original publication⁵ describing the database. The original database project utilized differing procedures for different types of cast iron. This paper only addresses ductile iron, so only the procedures used for ductile iron are contained herein. Furthermore, operative ASTM test methods are only contained in the original publication.

Chemical Analysis and Metallography

An analysis of alloy chemistry was conducted on each of two samples extracted from randomly selected castings from each group of test materials. The concentrations of most elements were determined using optical emission spectrometry (OES) except carbon, nitrogen, and sulfur, which were determined using combustometric methods. Copper and magnesium were determined by atomic absorption (AA) spectrophotometry.

Typically, three metallographic specimens were prepared and analyzed for each test material. Transverse cross-sections were extracted from the gage section of tensile bars, mounted and polished using standard metallographic techniques. The samples were final polished using colloidal silica media with a 0.05 um particle size. The samples were photographed at two or three appropriate magnifications in the as-polished condition and after etching with 2% Nital. Automated image analysis was performed to measure nodularity, nodule count, and ferrite/pearlite content.



Figure 1. Engineering and true stress-strain plots and properties from Reference 4.

Hardness and Tensile Testing

Brinell hardness measurements were conducted on the shoulders of fractured tensile bars using a 3000 kg load. Tensile specimens were tested to determine standard tensile properties [i.e., 0.2% offset yield strength (S_y or YS), ultimate tensile strength (S_u or UTS), total elongation (%El), and reduction in area (%RA)]. Additionally, the monotonic tensile stress-strain response was characterized for a subset of the specimens in terms of elastic modulus and/or Poisson's ratio as well as monotonic strength coefficient (K) and monotonic strain hardening exponent (n). Figures 1 and 2 show the standard tensile properties and monotonic stress-strain constants, respectively.⁴

Room temperature tensile tests were conducted on standard round tensile specimens with 0.5 inch (13 mm) gauge diameter and 2.0 inch (50 mm) gauge length. In isolated cases, proportional subsize tension specimens were used when necessitated by material size constraints.

Strain gauges were employed for precision modulus and Poisson's ratio determinations and an extensometer was employed for determining K and n. For standard tension tests, an initial strain-rate of 0.3% per minute was imposed until the yield strength was exceeded; after yield, the strain-rate was increased to 10% per minute. The total elongation was measured by reassembly of the samples and physical measurement of the final gauge length. For elastic modulus or Poisson's ratio determination, a smaller initial strain rate of 0.2% per minute was used.

Fatigue Testing

All fatigue test specimens were machined with low stress procedures and longitudinally polished. The specimens had a 0.314 inch (8 mm) gage diameter, 0.628 inch (16 mm) gage length, 2.51 inch (64 mm) blending radius, and 0.590 inch (15 mm) diameter grip ends.



Figure 2. Logarithmic true stress-plastic strain plot with monotonic stress-strain constants from Reference 4.

Strain-controlled fatigue tests were conducted at room temperature over the range from about 100 cycles to 1,000,000 cycles. Generally, three specimens were tested at each of six to seven strain levels, and all tests were performed using a triangular waveform at a constant strain rate of 2% per second.

Specimens with anticipated fatigue lives in excess of 500,000 cycles were switched to stress-controlled testing after 100,000 cycles and thereafter were tested at a constant frequency of 15 cycles/second. Tests were continued until a 20% drop in stress range (\sim 50% load drop in cast iron) occurred, or until 5,000,000 cycles were reached without failure. Strain and load data were recorded during the tests.

Fatigue Data Analysis

This section of the paper summarizes the strain-life fatigue data analysis that was conducted in the database development effort.⁷ The reader is directed to other references for further details.^{3,4,7,8}

A fundamental step in the strain-life analysis of cyclic property data is the decomposition of the total cyclic strain amplitude ($\Delta \varepsilon_{t}/2$) or ($\Delta \varepsilon/2$) into its component strains, i.e., plastic strain amplitude ($\Delta \varepsilon_{p}/2$) and elastic strain amplitude ($\Delta \varepsilon_{p}/2$) according to the equation:

$$(\Delta \varepsilon/2) = (\Delta \varepsilon_{o}/2) + (\Delta \varepsilon_{n}/2)$$
 Eqn. 1

In practice, the elastic component of strain is determined by dividing the stress amplitude ($\Delta\sigma/2$) at specimen half-life (0.5N_f) by the elastic modulus (E). The plastic strain amplitude is then given as the difference between the total strain amplitude and the elastic strain amplitude. Figure 3 shows the definition of the stress and strain ranges, which are twice their amplitude counterparts. The plastic strain data determined by this method are used in the calculation of both the strain/stress-life and cyclic stress-strain constants.

Using data obtained from each fatigue specimen at half-life, the cyclic stress-strain constants n' (cyclic strain hardening exponent) and K' (cyclic strength coefficient) were determined by regressing the stress amplitude ($\Delta\sigma/2$) versus plastic strain amplitude ($\Delta\epsilon_p/2$) in logarithmic coordinates. Figure 4 shows the cyclic stress-strain constants, which are related as follows:

$$\Delta \sigma/2 = K' (\Delta \varepsilon_2/2)^{n'}$$
 Eqn. 2

The cyclic stress-strain constants K' and n' were also calculated directly from the strain-life constants.

The cyclic stress-life and strain-life constants $\sigma'_{f,} b, \epsilon'_{f,}$ and c were calculated as follows:

$$\Delta \sigma/2 = \sigma'_{f} (2N_{f})^{b}$$
 Eqn. 3

Where:

 $\Delta\sigma/2$ = true stress amplitude 2N_c = reversals to failure (1

 $2N_{f}$ = reversals to failure (1 reversal = $\frac{1}{2}$ cycle) σ_{c}^{*} = fatigue strength coefficient

Eqn. 4

b = fatigue strength exponent

as shown in Figure 5; and

$$\varepsilon_{\rm p}/2 = \varepsilon_{\rm f}'(2N_{\rm f})^{\rm c}$$

Where:

 Δs

 $\Delta \varepsilon_{\rm p}/2$ = plastic strain amplitude

 $2\dot{N}_{f}$ = reversals to failure

 ε'_{f} = fatigue ductility coefficient

c = fatigue ductility exponent

as shown in Figure 6.



Figure 3. Stable hysteresis loop at half-life, showing the definitions of stress, total strain, elastic strain, and plastic strain ranges from Reference 4.



Figure 4. Logarithmic cyclic stress-plastic strain plot with cyclic stress-strain constants from Reference 4.

As discussed previously, the total strain is the sum of the elastic and plastic strains. In terms of strain amplitude,

$$\Delta \varepsilon/2 = \Delta \varepsilon_{e}/2 + \Delta \varepsilon_{p}/2$$
 Eqn. 5

The elastic term can be written as:

$$\Delta \varepsilon_{o}/2 = \Delta \sigma/2E$$
 Eqn. 6

We can now state this in terms of reversals to failure:

$$\Delta \varepsilon_{e}/2 = (\sigma'_{f}/E) \times (2N_{f})^{b}$$
 Eqn. 7

The plastic term is:

$$\Delta \varepsilon_{\ell}/2 = \varepsilon_{\ell}^{\prime} (2N_{\ell})^{c}$$
 Eqn. 8

The total strain can now be rewritten:

$$\Delta \varepsilon/2 = (\sigma_{\ell}' E) \times (2N_{\ell})^{b} + \varepsilon_{\ell}' (2N_{\ell})^{c}$$
 Eqn. 9

This last equation is the basis of the strain-life method and is termed the strain-life relation, as shown in Figure 7.

It is sometimes useful to calculate the transition fatigue life (N_t) , which is one definition for the fatigue life that is the boundary between low and high cycle fatigue. By replacing N_t for N_f in the above equations for elastic strain amplitude and plastic strain amplitude, the following equation is obtained:

$$N_{t} = \frac{1}{2} (\epsilon_{f}^{*} E / \sigma_{f}^{*})^{1/(b-c)}$$
 Eqn. 10

The cyclic stress-strain constants K' and n' were also calculated directly from the strain-life and stress-life constants as follows:

$$\begin{array}{ll} \mathbf{K}^{'} &= \sigma'_{\mathbf{f}} / (\epsilon'_{\mathbf{f}})^n \\ \mathbf{n}^{'} &= \mathbf{b} / \mathbf{c} \end{array}$$

Thus, the database contains K' and n' constants that were calculated by two methods, i.e., direct log-linear regression of the stress and plastic strain amplitudes and simple manipulation with the above two equations. Only the K' and n' data obtained from the regression analyses will be discussed in this paper.

Results and Discussion

Database Contents

The database has a directory structure that is summarized in the report, and other computer files. A typical database page for the composition and microstructure austempered ductile iron is shown in Figure 8. A typical database page with monotonic properties, strain-life fatigue constants and curves is shown in Figure 9. These database pages show average values, but the raw fatigue data and micrographs are also contained in the database. The turquoise-shaded regions are contributed datasets.



Figure 5. Logarithmic stress amplitude versus reversals plot from Reference 4.



Figure 6. Logarithmic plastic strain amplitude versus reversals plot from Reference 4.



Figure 7. Logarithmic total strain amplitude versus reversals plot from Reference 4, showing the addition of the logarithmic elastic strain and plastic strain versus reversals lines.

							Element (wt%)									Matrix Composi			(ASTM A-247)	Nodule Size Distribution (%)		Nodule Count (n	Nodularity (%		Material Code (Fold	Cast Section S		Material Condi	Grade: ASTM A 897	Iron Type
Sn	Ce	Ξ	Mg	Cu	AI	Mo	N	Cr	Mn	Si	P	s	C.E. = %C + 1/3(%SI)	c		tion		4	5	9	7	ım ⁻²)	6)		er Title)	ize		ion	'A 897M	
																Austenite & Acicular Ferrite	Substantially									ASTM Standard				
0.005	0.0056	0.01	0.056	0.54	0.12	<0.01	0.29	0.03	0.27	2.52	0.012	0.016	4.45	3.61		Acicular Ferrite	Austenite &	-	33	48	19	112	94		Grd 1 25mm	Y-block	25 mm	Austempered	125-80-10 [850-550-10]	
		-	-		-			•	-	-				•		Acicular Ferrite	Austenite &			-		-			Grd 1 (lit)	section	~25 mm	Austempered		
															Chemi	Austenite & Acicular Ferrite	Substantially							Micros		ASTM Standard				
0.005	0.0056	0.01	0.056	0.54	0.12	<0.01	0.29	0.03	0.27	2.52	0.012	0.016	4.45	3.61	cal Analysis	Acicular Ferrite	Austenite &	<u> </u>	33	48	19	112	94	tructure	Grd 2 25mm	Y-block	25 mm	Austempered	150-100-7 [1050-700-7]	
		•							•	•						Acicular Ferrite	Austenite &			•		•	•		Grd 2 (contrb)	section	~25 mm	Austempered		Aus
		·														Austenite & Acicular Ferrite	Substantially			•	×					Standard	ACTM			tempered Ducti
0.005	0.0056	0.01	0.056	0.54	0.12	<0.01	0.29	0.03	0.27	2.52	0.012	0.016	4.45	3.61		Acicular Ferrite	Austenite &	-	33	48	19	112	94		Grd 3 25mm	Y-block	25 mm	Austempered	175-125-04 [1200-850-04]	le Iron
	•	•		•						•	•			•		Acicular Ferrite	Austenite &			•		•	•		Grd 3 (contrb)	section	~25 mm	Austempere d		
•		•		•								•		•		Austenite & Acicular Ferrite	Substantially			-						ASTM Standard				
0.005	0.0056	0.01	0.056	0.54	0.12	<0.01	0.29	0.03	0.27	2.52	0.012	0.016	4.45	3.61		Acicular Ferrite	Austenite &	-	33	48	19	112	94		Grd 4 25mm	Y-block	25 mm	Austempered	200-155-01 [1400-1100-01]	
																Acicular Ferrite	Austenite &		•		•				Grd 4 (contrb	section	~25 mm	Austempered		

Figure 8. Typical database page with composition and quantitative microstructural results. This figure shows results for four tested grades and the contributed dataset shaded in gray.

Iron Type						Auste	mpered Ductil	e Iron				
Grade: ASTM A 807/A 807/M		125-80-10			150-100-7			175-125-04			200-155-01	
Giade. Ad Im A 03/14 03/14		[850-550-10]			[1050-700-7]			[1200-850-04]			1400-1100-01]	
Material Condition		Austempered	Austempered		Austempered	Austempered		Austempered	Austempere d		Austempered	Austempered
Cast Section Size	ASTM Standard	25 mm Y-block	~25 mm section	ASTM Standard	25 mm Y-block	~25 mm section	ASTM Standard	25 mm Y-block	~25 mm section	ASTM Standard	25 mm Y-block	~25 mm section
Material Code (Folder Title)		Grd 1 25mm	Grd 1 (lit)		Grd 2 25mm	Grd 2 (contrb)		Grd 3 25mm	Grd 3 (contrb)		Grd 4 25mm	Grd 4 (contrb)
			Monot	onic Properties	(Hardness and	Tension)						
Brinell Hardness (HBW3000)	269 - 321 ¹	304	302	302 - 363	341	340	341 - 444	390	387	388 - 477 ¹	441	418
Ultimate Tensile Strength (S _U), ksi (MPa)	>125 ksi	153 (1060)	140 (966)	> 150 ksi	168 (1160)	165 (1140)	> 175 ksi	197 (1360)	190 (1310)	> 200 ksi	225 (1550)	220 (1520)
Tensile 0.2% Offset Yield Strength (S _{ys}), ksi (MPa)	> 80 ksi	109 (752)	110 (759)	> 100 ksi	126 (869)	130 (897)	> 125 ksi	157 (1080)	160 (1100)	> 155 ksi	180 (1240)	180 (1240)
Percent Elongation (%)	> 10 %	13.3	11	> 7.0 %	13	10	> 4.0 %	6.7	7	> 1.0 %	3.1	5
Percent Reduction in Area (%)	•	12	10		13	6		8.6	9		2.9	4
Strain Hardening Exponent (n)	•	0.16	0.143		0.124			0.085	•		0.148	•
Strength Coefficient (K), ksi (MPa)		231 (1600)	218 (1500)		238 (1640)			265 (1830)			386 (2660)	
Elastic Modulus (E) ksi /MDa)		23800	23600		23600	23200		23100	22900		23300	22600
LIASUU MUUUUS (L), NSI (MF A)	•	(164000)	(163000)	•	(163000)	(160000)	•	(159000)	(158000)	•	(161000)	(156000)
Poisson's Ratio (μ)	-	0.275	0.25		0.262	0.25	×	0.278	0.25		0.27	0.25
				Cyclic Stress-	Strain Propertie	s						
Cyclic Strain Hardening Exponent (n')		0.125	0.1252		0.132	0.1376		0.116	0.1465		0.130	0.1600
Cyclic Strength Coefficient (K'), ksi (MPa)		235 (1620)	253 (1740)		279 (1920)			322 (2220)			467 (3120)	•
				Strain-life	Properties							
Fatigue Ductility Coefficient (c' _f)		0.5255	0.1150		0.3452	0.1780		0.3951	0.3960		0.2323	0.4880
Fatigue Ductility Exponent (c)	•	-0.7391	-0.5940	•	-0.6970	-0.6280		-0.7855	-0.7520		-0.8224	-0.8480
Fatigue Strength Coefficient (σ' _f), ksi (MPa)		208.6 (1440)	211 (1460)		253.4 (1750)	394 (2720)		400.3 (2760)	450 (3100)		439 (3030)	728 (5020)
Fatigue Strength Exponent (b)		-0.0864	-0.0900		-0.0985	-0.1460	×	-0.1371	-0.1600		-0.1459	-0.2050
Footnotes												

These hardness ranges are only typical and are not ASTM requirements.
 The gray shaded columns denote data taken from literature
 The blue shaded columns denote contributed data. Although the strain-rates and waveforms used to generate the contributed data may differ from those used in tests conducted at CRS, the tests were reportedly conducted in accordance with ASTM

(a)

properties and fatigue constants for the four ADI grades tested for the database, and the two contributed datasets shaded in turquoise. (b) This portion of the figure shows the individual strain-life plots and regression results for one of the ADI grades. Figure 9. Typical database page with average monotonic properties and fatigue information. (a) This portion of the figure contains the average monotonic



Figure 9. Typical database page with average monotonic properties and fatigue information. (a) This portion of the figure contains the average monotonic properties and fatigue constants for the four ADI grades tested for the database, and the two contributed datasets shaded in turquoise. (b) This portion of the figure shows the individual strain-life plots and regression results for one of the ADI grades.

Data Selected for Analysis in this Paper

Some of the elemental contents and quantitative microstructural constituents are summarized in Table 2 for the eight selected conditions from Table 1. All subsequent figure numbers and legends in this paper are ordered with respect to the particular column that the condition occupies in Table 2. The Table 2 results were chosen to display the most significant data for each of the selected conditions. Figures 10 to 17 each contain one example micrograph in the etched condition to illustrate each of the eight selected conditions.

Table 2. Abbreviated Elemental Contents and Microstructural Results for Selected Database Condit	ions
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Grade	: ASTM A 536 [SAE J434]	6	0-40-18 [D40	18]	65-45-12 [D4512]	1	100-70-03 [D70	03]	120-90-02		
Ma	aterial Condition	As-cast	Subcritical Anneal	Full Anneal	As-cast	As-cast	Normalized	Normalized	Quenched & Tempered		
Ca	ast Section Size	25 mm Keel-block	25 mm Keel-block	25 mm Y-block	25 mm Keel-block	25 mm Keel-block	25 mm Y-block	76 mm Y-block	25 mm Keel-block		
Materia	al Code (Folder Title)	4018-AC- 25mm	4018- ScAn- 25mm	4018-FAn- 25mm	4512-AC- 25mm	7003-AC- 25mm	7003-N- 25mm	7003-N- 76mm	9002-QT- 25mm		
Microstructure											
	Nodularity (%)	93	95	93	92	94	95	96	94		
Noc	dule Count (mm ⁻²)	264	246	202	112	271	153	40	299		
Matrix Composition		81.3% Ferrite 18.7% Pearlite	Fully Ferritic	Fully Ferritic	54% Ferrite 46% Pearlite	8.7% Ferrite 91.3% Pearlite	8% Ferrite 92% Pearlite	1.7% Ferrite 98.3% Pearlite	Tempered Martensite		
				Chemical A	nalysis						
	С	3.61	3.61	3.65	3.84	3.59	3.63	3.65	3.59		
	C.E. = %C + 1/3(%SI)	4.50	4.50	4.53	4.61	4.35	4.50	4.52	4.35		
	S	0.008	0.008	0.009	0.008	0.008	0.01	0.007	0.008		
Elomont	Р	0.026	0.026	0.016	0.02	0.02	0.023	0.026	0.02		
(wt%)	Si	2.66	2.66	2.63	2.31	2.29	2.6	2.6	2.29		
	Mn	0.16	0.16	0.21	0.17	0.22	0.22	0.3	0.22		
	Cu	0.25	0.25	0.15	0.32	0.85	0.23	0.6	0.85		
	Mg	0.036	0.036	0.036	0.029	0.037	0.047	0.049	0.037		
	Се	0.0058	0.0058	0.0072	0.0038	0.0056	0.0035	0.0049	0.0056		



Figure 10. This optical micrograph corresponds to grade 60-40-18 in the as-cast condition. The original magnification is 200X.



Figure 11. This optical micrograph corresponds to grade 60-40-18 in the subcritically annealed condition. The original magnification is 250X.



Figure 12. This optical micrograph corresponds to grade 60-40-18 in the fully annealed condition. The original magnification is 250X.



Figure 13. This optical micrograph corresponds to grade 65-45-12 in the as-cast condition. The original magnification is 250X.



Figure 14. This optical micrograph corresponds to grade 100-70-13 in the as-cast condition. The original magnification is 250X.



Figure 15. This optical micrograph corresponds to the 25 mm section size of grade 100-70-13 in the normalized condition. The original magnification is 250X.



Figure 16. This optical micrograph corresponds to the 76 mm section size of grade 100-70-13 in the normalized condition. The original magnification is 250X.



Figure 17. This optical micrograph corresponds to grade 120-90-02 in the quenched and tempered condition. The original magnification is 500X.

Compositions and Microstructures

Table 2 shows that all the as-cast and heat treated ductile cast irons have typical compositions for their corresponding $ASTM^9$ or SAE^{10} grades, all have high nodularity (greater than 90%), and almost all have standard nodule counts between 100 to 300 mm⁻². Due to its heavier section size, the 76 mm section size for the 100-70-03 grade has lower nodule count than the other seven conditions.

The Table 2 results were chosen to display the most significant data for each of the selected conditions. Figures 10 to 17 each contain one example micrograph selected for each of the eight conditions.

The microstructures in the grades obeyed expectations as follows:



- 1. The ferrite content in the as-cast grades decreased with increasing strength and decreasing ductility in the first three grades listed in Table 2, as also shown in Figures 10, 13 and 14.
- 2. The ferrite contents in Table 2 and the micrographs in Figures 10 to 12 show that the annealing treatments converted the as-cast 60-40-18 grade to all ferrite, as expected.
- 3. For as-cast 100-70-03, Figure 14 shows that the ferrite surrounded nodules, and this is commonly called bulls-eye ferrite. After normalizing, Figures 15 and 16 show that the ferrite is more randomly distributed throughout the matrix.
- 4. For grade 120-90-02, Figure 17 shows that the ferrite-pearlite microstructure converted to tempered martensite after quenching and tempering (Q&T).

Figure 18. The yield strength and elongation of the ferrite and pearlite microstructures obeyed a quadratic relationship. The open points represent the heat treatment conditions and the closed points represent the as-cast condition. The tempered martensite microstructure had greater strength than the locus of points for the ferrite and pearlite grades.

Grade: ASTM A 536 [SAE J434]	(60-40-18 [D4018	3]	65-45-12 [D4512]	1	100-70-03 [D700	3]	120-90-02
Material Condition	As-cast	Subcritical Anneal	Full Anneal	As-cast	As-cast	Normalized	Normalized	Quenched & Tempered
Cast Section Size	25 mm Keel-block	25 mm Keel-block	25 mm Y-block	25 mm Keel-block	25 mm Keel-block	25 mm Y-block	76 mm Y-block	25 mm Keel-block
Material Code (Folder Title)	4018-AC- 25mm	4018-ScAn- 25mm	4018-FAn- 25mm	4512-AC- 25mm	7003-AC- 25mm	7003-N- 25mm	7003-N- 76mm	9002-QT- 25mm
		Monotonic P	roperties (Haro	ness and Te	nsion)		•	
Brinell Hardness (HBW3000)	161	154	150	186	259	269	285	296
Ultimate Tensile Strength (S _u), ksi (MPa)	69.6 (480)	66.5 (459)	64.2 (443)	84 (579)	121.3 (836)	128 (883)	129.5 (893)	133.7 (922)
0.2% Offset Yield Strength (S _{ys}), ksi (MPa)	46.9 (323)	46.4 (320)	41.7 (288)	49.3 (340)	70.6 (487)	74.4 (513)	79.6 (549)	108.8 (750)
Percent Elongation (%)	21	21	25	13	8.1	8.2	4.7	6.8
Percent Reduction in Area (%)	24	26	30	11	7.1	7.2	3.97	7.2
Strain Hardening Exponent (n)	0.175	0.16	0.18	0.216	-	0.252	0.279	0.119
Strength Coefficient (K), ksi (MPa)	114 (786)	107 (738)	103 (710)	154 (1060)	-	269 (1860)	324 (2230)	200 (1380)
Elastic Modulus (E), ksi (MPa)	25000 (172000)	25100 (173000)	24000 (165000)	25100 (173000)	24900 (172000)	24800 (171000)	25700 (177000)	24600 (170000)
Poisson's Ratio (μ)	0.289	0.287	0.282	0.288	-	0.287	0.293	0.278

Monotonic Mechanical Properties

Table 3 shows the monotonic mechanical properties that were obtained for the selected grades. Figure 18 shows a plot of yield strength versus elongation for all eight selected conditions. Selected mechanical properties are also plotted versus ferrite content in Figures 19 and 20 for the three ascast conditions.

The monotonic mechanical properties in the as-cast condition obeyed expectations as follows:

- 1. Figure 18 shows that the yield strengths of the ferritic, ferritic-pearlitic, and pearlitic irons decreased with increasing elongation.
- 2. Figure 19 shows that the hardness, yield strength and tensile strength increased with decreasing ferrite content in the as-cast conditions.
- 3. Figure 19 shows that the ductility, as measured by both elongation and reduction in area, increased with increasing ferrite content in the ascast conditions.

The monotonic mechanical properties were altered with heat treatment for the two grades that were evaluated, but less predictably. The following heat treatment trends were observed:

- 1. Table 3 shows that the tensile strength of as-cast grade 60-40-18 decreased slightly with subcritical annealing and decreased significantly with full annealing.
- 2. Table 3 shows that the yield strength of as-cast grade 60-40-18 decreased slightly with subcritical annealing and decreased significantly with full annealing.
- 3. Table 3 shows that the ductility of as-cast grade 60-40-18 hardly changed with subcritical annealing; however, significant increases in ductility were obtained with full annealing.
- 4. Table 3 shows that the yield and tensile strengths of the normalized 100-70-03 grade increased significantly with respect to the as-cast 100-70-03 grade in both the 25 mm and 76 mm section sizes. (In fact, the 100-70-03 properties are not generally obtainable in a heavy section size without normalizing.)

- 5. Table 3 shows that ductility of the 25 mm as-cast samples of the 100-70-03 grade did not change when they were normalized.
- 6. Table 3 shows that the ductility of the normalized 76 mm samples was significantly lower than the 25 mm samples of the 100-70-03.
- Figure 18 shows that quenched and tempered (Q&T) grade 120-90-02 had a significantly higher yield strength at the same ductility of the pearlitic 100-70-03 grade.

No significant change in the elastic properties (modulus and Poisson's ratio) were observed for any of the conditions.



Figure 19. The hardness and strength decreased with increasing ferrite content for the as-cast conditions.



Figure 20. The ductility increased with increasing ferrite content for the as-cast conditions.

Strain-Life Fatigue Properties

10.0%

Total Strain Amplitude (%)

Table 4 shows the cyclic fatigue constants that were obtained for the selected grades. Note that it is both difficult and not recommended to directly compare materials based on the constants alone. Similarities and differences among materials are best compared by strain-life plots, as discussed below.

Figure 21 shows a plot of the total strain versus fatigue cycles for four selected grades. In the plot legend, the grade is identified with the ASTM designation, but the actual UTS-YS-El values from Table 3 and ferrite contents from Table 2 are shown parenthetically. The plotted curves in Figure 21 are the total strain-life hyperbolas predicted from the results in Table 4. Actual fatigue data points are shown for three grades. Predicted fatigue lives are shown in Table 5.

The cyclic properties of the grades obeyed expectations as follows:

- 1. Figure 21 and Table 5 show that the higher strength grades had lower fatigue lives in the low cycle regime and higher fatigue lives in the high cycle regime.
- 2. Figure 21 and Table 5 show that the higher ductility grades had higher fatigue lives in the low cycle regime and lower fatigue lives in the high cycle regime.
- 3. These predicted trends were observable, but only showed significance for the extremes, i.e., the differences in fatigue lives were only significantly greater than the range of the data for the two extreme conditions (the 60-40-18 and 120-90-02 grades).

60-40-18 Raw data

60-40-18 (70-47-21) As-Cast (100%f)

Figure 21. Total strain-life fatigue curves are shown as a function of grade and display the expected trends. The points are actual data for three selected grades and the curves are the predicted lives from constants. The higher strength grades exhibit greater lives at high cycles and the higher ductility (and lower strength) grades exhibit greater lives in the low cycle regime. The values in parentheses in the legend are the measured monotonic properties and ferrite contents.

Grade: ASTM A 536 [SAE J434]	60)-40-18 [D401	8]	65-45-12 [D4512]		100-70-03 [D70	03]	120-90-02					
Material Condition	As-cast	Subcritical Anneal	Full Anneal	As-cast	As-cast	Normalized	Normalized	Quenched & Tempered					
Cast Section Size	25 mm Keel- block	25 mm Keel-block	25 mm Y-block	25 mm Keel-block	25 mm Keel-block	25 mm Y-block	76 mm Y-block	25 mm Keel-block					
Material Code (Folder Title)	4018- AC- 25mm	4018- ScAn- 25mm	4018- FAn- 25mm	65-45-12 [D4512] uil heal As-cast A mm 25 mm keel-block 2 lock 4512-AC- 25mm 70 rain-life Method (ASTM 653 0.094 0 553 0.094 0 0 662 0.4482 0 602 -0.6483 -0 0.5 109.9 1 (757) 0 0 385	7003-AC- 25mm	7003-N- 25mm	7003-N- 76mm	9002-QT- 25mm					
Cyclic	Cyclic Stress-Strain Properties: Strain-life Method (ASTM E606; SAE J1099)												
Cyclic Strain Hardening Exponent (n')	0.073	0.074	0.053	0.094	0.087	0.105	0.122	0.091					
Cyclic Strength Coefficient (K'), ksi (MPa)	98.8 (681)	97.8 (674)	85.2 (587)	115 (793)	137 (945)	156 (1080)	178 (1230)	172 (1190)					
S	train-life P	roperties: St	rain-life Me	ethod (ASTM	E606; SAE .	J1099)		-					
Fatigue Ductility Coefficient (ε' _f)	0.5148	0.6016	0.9632	0.4482	0.5855	0.6648	0.2729	0.3694					
Fatigue Ductility Exponent (c)	-0.6827	-0.7177	-0.7602	-0.6483	-0.7198	-0.6679	-0.6395	-0.7035					
Fatigue Strength Coefficient (o'f), ksi	102.1	103.3	99.5	109.9	134.2	163.8	164.8	191.7					
(MPa)	(704)	(712)	(686)	(757)	(926)	(1130)	(1140)	(1320)					
Fatigue Strength Exponent (b)	-0.0596	-0.0653	-0.0606	-0.0643	-0.0669	-0.0880	-0.0907	-0.0909					
Transition Fatigue Life (Nt), cycles	1227	1041	1200	385	658	1420	461	274					

Table 4. Low Cycle Fatigue Constants for Selected Database Conditions

Table 5 contains predictions and Figure 22 shows a plot of the total strain versus predicted fatigue cycles for the two heat treated ferritic-pearlitic grades (60-40-18 and 100-70-03), which had corresponding as-cast conditions. The cyclic properties of the heat treated grades generally obeyed expectations as follows:

- 1. In most conditions, the heat treatments produced greater low cycle fatigue lives for high ductility conditions
- 2. Although the high cycle fatigue resistance was affected less by heat treatment, slightly increased high cycle fatigue lives were obtained with higher strength conditions.
- 3. The normalized 25 mm samples of grade 100-70-03 exhibited significantly greater monotonic strength as well as fatigue resistance in both the low and high cycle regime.

Table 5 contains predictions and Figures 23 and 24 respectively show predicted stress-life and plastic strain-life fatigue curves as a function of grade. Table 4 shows no observable trends in transition fatigue life.

Summary and Conclusions

- 1. This paper reviewed the contents and architecture of the AFS strain-life fatigue database for cast iron.
- 2. As expected, the monotonic
 - a. Yield strengths of the ferritic, ferritic-pearlitic, and pearlitic irons decreased with increasing elongation,
 - b. Hardness, yield strength and tensile strength increased with decreasing ferrite content.
 - c. Ductility increased with increasing ferrite content.
 - d. Elastic modulus and Poisson's ratio were virtually identical for all the tested conditions.
- 3. The monotonic mechanical properties were altered with heat treatment for the 60-40-18 and 100-70-03 grades as follows:
 - a. The yield and tensile strengths of as-cast grade 60-40-18 decreased slightly with subcritical annealing and decreased significantly with full annealing.
 - b. The ductility of as-cast grade 60-40-18 hardly changed with subcritical annealing; however, significant increases in ductility were obtained with full annealing.

							•			
Material & Allo	у		60-40- 18 (70- 47-21) As-Cast (100%f)	60-40-18 (67-46- 21) Subcritical Anneal (100%f)	60-40- 18 (64- 42-25) Full Anneal (81%f)	65-45- 12 (84- 49-13) As-Cast (54%f)	100-70- 03 (121- 71-08) As-Cast (9%f)	100-70-03 (128-74- 08) Normalized (8%f)	100-70-03 (128-80-05) Normalized Heavy Section (2%f)	120-90-02 (134-109-07) Q&T (100%TM)
Condition Iden	tifier	Cycles	4018- AC- 25mm	4018- ScAn- 25mm	4018- FAn- 25mm	4512- AC- 25mm	7003- AC- 25mm	7003-N- 25mm	7003-N- 76mm	9002-QT- 25mm
		100	511	504	498	538	650	709	705	815
Stress		1,000	444	433	433	464	557	579	572	661
Amplitude,	Δσ / 2	10,000	386	373	376	400	477	473	464	537
MPa		100,000	336	321	327	433 464 557 579 572 376 400 477 473 464 327 345 409 386 377 285 298 351 315 306 0.302% 0.311% 0.378% 0.415% 0.398% 0.262% 0.268% 0.324% 0.339% 0.323% 0.228% 0.231% 0.276% 0.262% 0.262% 0.198% 0.200% 0.238% 0.226% 0.213% 0.173% 0.172% 0.204% 0.184% 0.173%	435			
		1,000,000	292	276	285	298	351	315	100-70-03 (128-80-05) Normalized Heavy Section (2%f) 7003-N- 76mm 705 572 464 377 306 0.398% 0.323% 0.262% 0.213% 0.213% 0.213% 0.2113% 0.0215% 0.2113% 0.0111% 0.0025% 1.320% 0.3311% 0.224% 0.175%	353
Elastic Strain Amplitude, %		100	0.297%	0.291%	0.302%	0.311%	0.378%	0.415%	0.398%	0.480%
		1,000	0.258%	0.251%	0.262%	0.268%	0.324%	0.339%	0.323%	0.389%
	$\Delta\epsilon_{e}/2$	10,000	0.225%	0.216%	0.228%	0.231%	0.278%	0.276%	0.262%	0.316%
		100,000	0.195%	0.185%	0.198%	0.200%	0.238%	0.226%	0.213%	0.256%
		1,000,000	0.170%	0.160%	0.173%	0.172%	0.204%	0.184%	0.173%	0.208%
		100	1.7158%	1.3423%	1.7158%	1.4445%	1.2919%	1.9312%	0.9215%	0.8886%
		1,000	0.2980%	0.2571%	0.2980%	0.3246%	0.2463%	0.4149%	0.2113%	0.1759%
Amplitude %	$\Delta \epsilon_{p} / 2$	10,000	0.0518%	0.0493%	0.0518%	0.0730%	0.0470%	0.0891%	0.0485%	0.0348%
, inpitude, it		100,000	0.0090%	0.0094%	0.0090%	0.311% 0.378% 0.415% 0.398% 0.268% 0.324% 0.339% 0.323% 0.231% 0.278% 0.276% 0.262% 0.200% 0.238% 0.226% 0.213% 0.172% 0.204% 0.184% 0.173% 1.4445% 1.2919% 1.9312% 0.9215% 0.3246% 0.2463% 0.4149% 0.2113% 0.0730% 0.0470% 0.0891% 0.0485% 0.0164% 0.0090% 0.0191% 0.0111% 0.0037% 0.0017% 0.0041% 0.0025%	0.0069%			
		1,000,000	0.0016%	0.0018%	0.0016%	0.0037%	0.0017%	0.0041%	Normalized Heavy Section (2%f) 7003-N- 76mm 705 572 464 377 306 0.398% 0.262% 0.213% 0.213% 0.213% 0.0485% 0.0111% 0.0025% 1.320% 0.311% 0.224% 0.175%	0.0014%
		100	2.013%	1.634%	2.017%	1.756%	1.670%	2.346%	1.320%	1.368%
o		1,000	0.556%	0.508%	0.560%	0.593%	0.570%	0.753%	0.535%	0.565%
Amplitude %	Δε _T /2	10,000	0.276%	0.265%	0.280%	0.304%	0.325%	0.366%	0.311%	0.350%
/		100,000	0.204%	0.195%	0.207%	6 0.311% 0.378% 0.415% 0.398 6 0.268% 0.324% 0.339% 0.323 6 0.231% 0.278% 0.276% 0.262 6 0.200% 0.238% 0.226% 0.213 6 0.200% 0.238% 0.226% 0.213 6 0.172% 0.204% 0.184% 0.173 6 0.172% 0.204% 0.184% 0.173 6 0.3246% 0.2463% 0.4149% 0.2113 76 0.3246% 0.2463% 0.4149% 0.2113 76 0.3246% 0.2463% 0.4149% 0.2113 76 0.0730% 0.0470% 0.0891% 0.0485 76 0.0164% 0.0090% 0.0191% 0.0111 76 0.0037% 0.0017% 0.0041% 0.0025 76 1.570% 2.346% 1.320 6 76 0.593% 0.570% 0.753% 0.535 76	0.224%	0.263%		
		1,000,000	0.171%	0.161%	0.174%	0.176%	0.206%	0.188%	0.175%	0.209%

Table 5. Predicted Fatigue Lives from $\Delta \varepsilon_{1}/2 = (\sigma'_{1}/E) (2N_{2})^{b} + (\varepsilon'_{2}) (2N_{2})^{c}$ for Selected Database Conditions

- c. The yield and tensile strengths of the normalized 100-70-03 grade increased significantly with respect to the as-cast 100-70-03 grade in both the 25 mm and 76 mm section sizes.
- d. The ductility of 25 mm as-cast samples of the 100-70-03 grade did not change when they were normalized.
- e. The ductility of the normalized 76 mm samples was significantly lower than the 25 mm samples of the 100-70-03.
- f. The quenched and tempered grade 120-90-02 had a significantly higher yield strength at the same ductility of the pearlitic 100-70-03 grade.
- 4. The cyclic properties of the grades obeyed expectations as follows:
 - a. As compared to the lower strength grades, the

higher strength grades had lower fatigue lives in the low cycle regime and higher fatigue lives in the high cycle regime.

- b. As compared to the lower ductility grades, the higher ductility grades had higher fatigue lives in the low cycle regime and lower fatigue lives in the high cycle regime.
- c. These predicted trends were observable, but only showed significance for the extremes, i.e., the differences in fatigue lives were only significantly greater than the range of the data for the two extreme conditions (the 60-40-18 and 120-90-02 grades).
- 5. No observable difference in transition (between the low and high cycle or plastic and elastic) fatigue life was observed.



Figure 22. Total strain-life fatigue curves are shown for the two heat treated ferritic-pearlitic grades with corresponding as-cast conditions. The points are actual data for three selected grades and the curves are the predicted lives from constants. The values in parentheses in the legend are the measured monotonic properties and ferrite contents. In most conditions, the heat treatments did not significantly affect the predicted fatigue life for the as-cast condition, although similar strength and ductility trends to those shown in Figure 21 were observed. It is noteworthy that the normalized 25 mm samples of grade 100-70-03 exhibited significantly greater monotonic strength and fatigue resistance in both the low and high cycle regime.

Figure 23. Predicted stress-life fatigue curves are shown as a function of grade and display the expected trends for all eight selected conditions. The higher strength grades exhibit greater lives at high cycles and decreased lives in the low cycle regime. The values in parentheses in the legend are the measured monotonic properties and ferrite contents.



Figure 24. Predicted plastic strain-life fatigue curves are shown as a function of grade and display the expected trends for all eight selected conditions. The higher ductility grades exhibit greater lives at low cycles and decreased lives in the high cycle regime. The values in parentheses in the legend are the measured monotonic properties and ferrite contents.

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